

*Molecular dynamics computer simulation
of scratch resistance testing of polymers:
visualization*

**Witold Brostow, Sameer Deshpande,
Travis Hilbig & Ricardo Simoes**

Polymer Bulletin

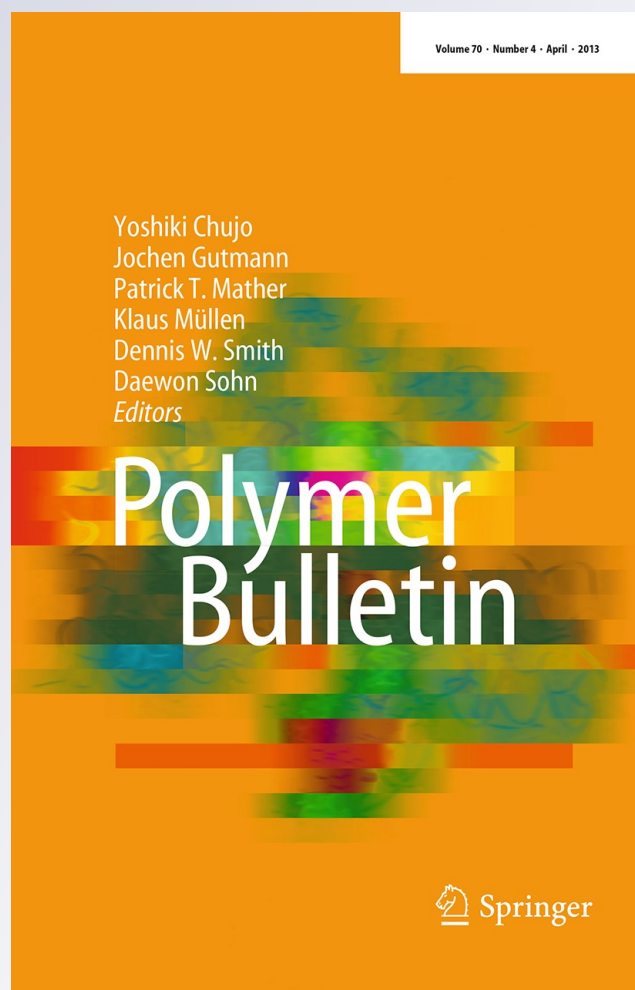
ISSN 0170-0839

Volume 70

Number 4

Polym. Bull. (2013) 70:1457-1464

DOI 10.1007/s00289-013-0949-5



Your article is protected by copyright and all rights are held exclusively by Springer-Verlag Berlin Heidelberg. This e-offprint is for personal use only and shall not be self-archived in electronic repositories. If you wish to self-archive your work, please use the accepted author's version for posting to your own website or your institution's repository. You may further deposit the accepted author's version on a funder's repository at a funder's request, provided it is not made publicly available until 12 months after publication.

Molecular dynamics computer simulation of scratch resistance testing of polymers: visualization

Witold Brostow · Sameer Deshpande · Travis Hilbig · Ricardo Simoes

Received: 17 July 2012 / Accepted: 1 February 2013 / Published online: 28 February 2013
© Springer-Verlag Berlin Heidelberg 2013

Abstract Experimental scratch resistance testing provides two numbers: the penetration depth R_p and the healing depth R_h . In molecular dynamics computer simulations, we create a material consisting of N statistical chain segments by polymerization; a reinforcing phase can be included. Then we simulate the movement of an indenter and response of the segments during X time steps. Each segment at each time step has three Cartesian coordinates of position and three of momentum. We describe methods of visualization of results based on a record of $6NX$ coordinates. We obtain a continuous dependence on time t of positions of each of

W. Brostow (✉) · S. Deshpande · T. Hilbig · R. Simoes
Department of Materials Science and Engineering and Department of Physics,
Laboratory of Advanced Polymers and Optimized Materials (LAPOM),
University of North Texas, 3940 North Elm Street, Denton, TX 76207, USA
e-mail: wbrostow@yahoo.com
URL: <http://www.unt.edu/LAPOM/>

S. Deshpande
e-mail: sameerd@mit.edu

T. Hilbig
e-mail: travishilbig@gmail.com

R. Simoes
e-mail: rsimoes@dep.uminho.pt; rsimoes@ipca.pt

S. Deshpande
Department of Mathematics, Massachusetts Institute of Technology,
Cambridge, MA 02139, USA

R. Simoes
Institute for Polymers and Composites, IPC/I3 N, University of Minho,
Campus de Azurem, 4800-058 Guimaraes, Portugal

R. Simoes
School of Technology, Polytechnic Institute of Cavado and Ave,
Campus do IPCA, 4750-810 Barcelos, Portugal

the segments on the path of the indenter. Scratch resistance at a given location can be connected to spatial structures of individual polymeric chains.

Keywords Molecular dynamics simulation · Scratch resistance · Polymer tribology

Introduction

The usage of polymer based materials (PBMs, polymers and polymer containing composites) grows at a fast rate. Most literature on PBMs deals with their mechanical properties, thermophysical properties or else with processing and rheology. There is much less literature on tribology of PBMs: their scratch resistance, wear and friction—important in a number of applications such as coatings in electronics.

The term tribology has been coined in 1966, in a British governmental report of a committee headed by Jost [1, 2]. In 1965, appeared an important book by Ernest Rabinowicz [3] which has its second edition 30 years later [4]. In 1978, appears a book by Czichos [5] who advocates a systems approach to tribology. Rabinowicz [4] says, “The Jost Report actually greatly underestimates the financial importance of tribology. The Report paid little attention to wear, which happens to be (from the economical point of view) the most significant tribological phenomenon”. For metals and polymers, we can mitigate scratching and wear and lower friction by the application of external lubricants, mostly liquid ones. Applying them to PBMs, we find that often swelling of the material takes place [6, 7]; the lubricant makes only the situation worse—so that other approaches have to be developed.

Computer simulations including Monte Carlo and molecular dynamics (MD) methods can provide a wealth of information and explain mechanisms of behavior inaccessible experimentally [8–14]. The situation described above behooves us to acquire *understanding of mechanisms of scratching* PBM surfaces. The key problem is obtaining pertinent information from a huge amount of results of each simulation run.

Molecular dynamics simulations of PBMs

We use the concept of statistical chain segments so that each particle in our simulations represents a number of real chain polymeric segments [15]. We shall deal with the creation of a material in the next section. MD simulations procedure has been described before; see for instance [9]. At the beginning of each time step updated velocities are calculated for each particle, in our situation as a function of the total force F acting on a given particle. F consists of forces created by intramolecular and intermolecular interactions, plus the external applied force. The new velocities are then scaled to maintain a constant temperature.

Simulations have been performed using the molecular dynamics method, initially developed by Alder and Wainwright [16] at the Lawrence Livermore Laboratory. Segments interact according to a set of pre-defined so-called Lennard-Jones

potentials [17] (originally defined by Mie [18]), which are defined as a function of the intersegmental distance. A detailed description of the simulation model has been provided before [19]. The employed simulation parameters and interaction potentials have been discussed earlier [20].

The problem is the amount of information resulting from each MD simulation run. The material consists of N statistical chain segments. To simulate experimental scratch resistance determination, we provide time for equilibration and then simulate the movement of an indenter applying force F on the surface and recording the response of the segments—for a total of X time steps. Each segment at each time step has three Cartesian coordinates of position and three of momentum along the Cartesian axes. The result is a record of $6NX$ stored coordinates. We describe in this paper the procedures of extracting meaningful information from such a record.

Creation of a polymer

Creation of a polymeric material on a computer has been also described before [9]. One starts on a lattice of particles, performs polymerization, and then performs equilibration of the material in function of the interaction potentials, internal inside the chains and external ones. The result is an off-lattice material, with a certain amount of free volume introduced during polymerization. We first need to see the material. Composites and nanocomposites of the type polymer plus reinforcement are growing in importance [21–29]. Therefore, we need the capability to place reinforcement at specific locations throughout the polymeric matrix. Simple shapes of reinforcing particles include spherical or cylindrical (fibers, carbon nanotubes). Placing small spherical particles is relatively easy. In Fig. 1, we show a more complicated case: a PBM consisting of a polymeric matrix plus cylindrical reinforcement placed at specific locations with respect to the material surface.

There is virtually an infinity of spatial structures of chains. We need to be able to track the chains individually—at least those which have one or more segments in the path of the indenter. Therefore, we provide different colors to the chains; see Fig. 2 for a neat polymer.

The amount of information in Fig. 2 is already large. We have the capability to focus on, say, two chains and see how at the local level they have responded to the deformation caused by the indenter; see Fig. 3. Note that Fig. 3 is in a 3D perspective, thus objects further away from the viewer look smaller.

Simulation of scratching

The scratching program, we have created, allows application of indenters of various size and shape. This is illustrated in Fig. 4. Segments are illustrated by spheres, those on which the indenter is actually exercising a force at a given time are marked red.

Both round and square indenters have been used in simulations thus far; actually, any shape could be defined for the indenter geometry. The indenter sizes currently considered range from a diameter of 2–7 statistical segments. On the one hand, we

Fig. 1 A polymer with a cylindrical reinforcement at various levels with respect to the surface

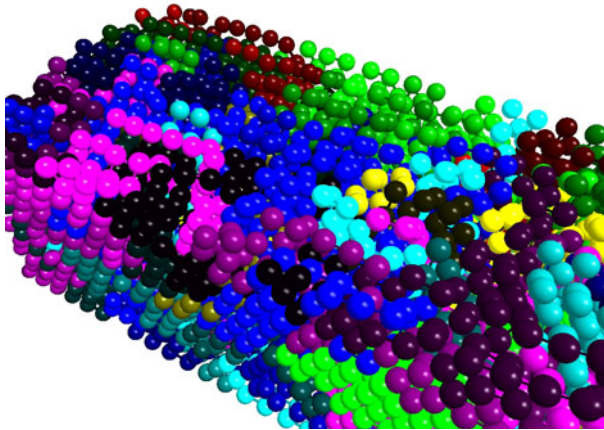
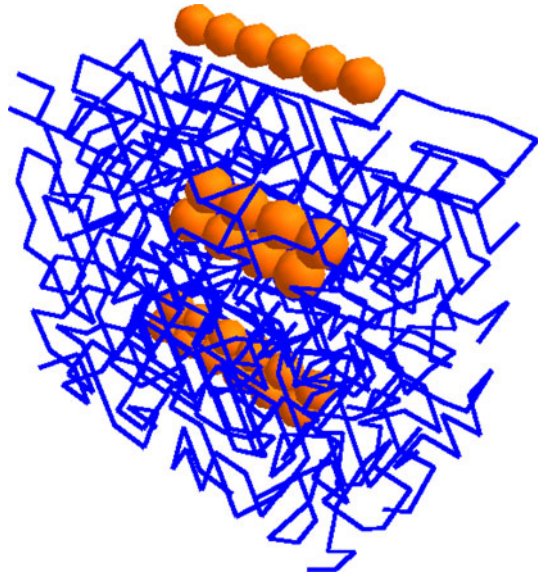


Fig. 2 A neat polymer with each chain marked by a different *color*

want to ensure that, at the lower end, the indenter always acts on several segments simultaneously. On the other hand, we defined that the indenter diameter should not extend more than half the width of the material—so as to prevent edge effects.

Analysis of simulation results

Again there is a problem of extracting pertinent information—such that could be connected among others to the spatial distribution of the chains as exemplified in Fig. 2. Experimental scratch resistance testing of polymers provides two numbers:

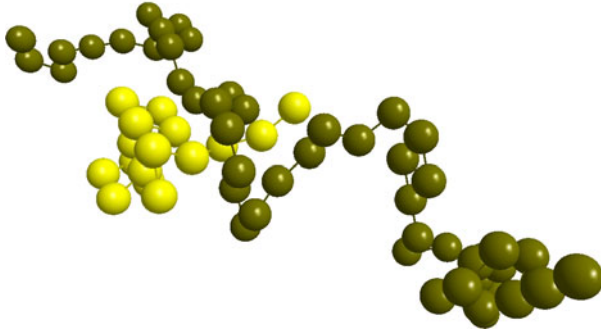


Fig. 3 Two chains only

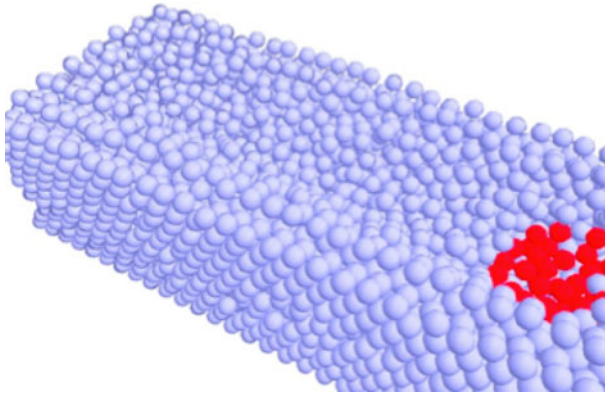


Fig. 4 View from the *top* for an indenter moving from *left to right*. The size and actual location of the indenter are marked by *red* segments. We see the groove already formed behind the track of the indenter

the instantaneous or penetration depth R_p and the recovery or healing depth R_h . We need to relate these numbers to MD results.

As previously stated, we obtain from the simulations 6NX coordinates. While the simulation is running, it is impossible to predict exactly which segments will be scratched subsequently; this depends on how the entire surface responds to the scratching thus far. Therefore, calculations and visualizations have to be done after the simulation has been terminated. The individual data files for each time step are also used by a 3D visualization application, using OpenGL rendering [30]; see Fig. 4. This application enables the user to freely pan and rotate the material in 3D, and to animate its behavior along time. OpenGL is a low-level graphics library specification. As OpenGL is not a programming language, but a library, it can be called from applications written in any of the major high-level programming languages. It makes available to the programmer a small set of geometric primitives such as points, lines, polygons, images, and bitmaps which can be rendered in a virtual 3D space and then visualized from any desired perspective. OpenGL provides a set of commands that allow the specification of geometric objects in two

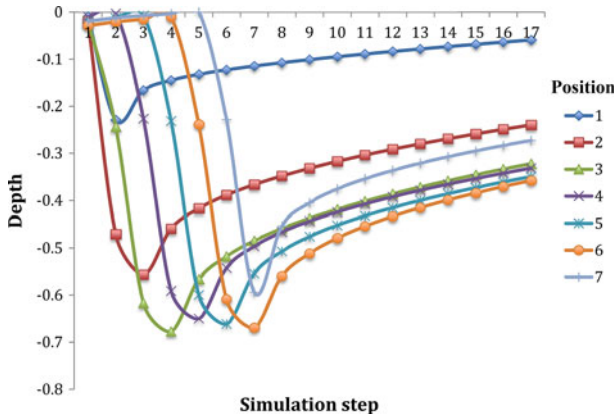


Fig. 5 Depth of groups of segments along the path of the indenter as a function of time t

or three dimensions, using the provided information along with commands that control how these objects are presented.

In the present work, the authors have coded their graphical application in Microsoft Visual C++ under a Windows 32-bit framework, and employed the OpenGL library to render each of the material elements (statistical segments, bonds, nanoparticles, and nanofibers).

The indenter hits several segments on the surface simultaneously; it is the combined response of those segments which is pertinent. The simulation progress is recorded periodically to status files, together with information on which segments were being scratched at that moment. After the simulation is completed, an application reads all these files, establishes the precise motion of the indenter along time and the respective scratched segments, and searches all the results files for the vertical position of each pertinent group of scratched segments. That information is then exported to a simple character delimited text file. A spreadsheet application can then open these files and produce plots such as that in Fig. 5. The diameter of the indenter is here comparable to the size of three segments.

We see in Fig. 5 that shortly after being ‘attacked’ by the indenter each segments reach a certain maximum depth. The average of these depths provides the penetration depth R_p , also seen in experiments. Since all polymers are viscoelastic, a recovery takes place—as it does in experiments. We see that each segment reaches a certain horizontal plateau—typically below its original position at the surface. The average of these plateaus is the residual or healing depth, R_h . In experiments one has defined the viscoelastic recovery f [31] as

$$f = [(R_p - R_h) 100\%]/R_p \tag{1}$$

One can also consider the brittleness B [32, 33]

$$B = 1/(E'\epsilon_b) \tag{2}$$

where E' is the storage modulus determined at 1.0 Hz by dynamic mechanical analysis [34], while ϵ_b is the elongation at break in tensile testing. There is a

relationship between f and B valid for a large variety of polymers and composites, namely [32, 33]

$$f = 30.6 + 67.1e^{-B/0.505} \quad (3)$$

To compare simulation results with experimental ones, we need to compare pairwise the values of R_p , R_h and the resulting viscoelastic recoveries f .

As said above, experiments provide only two numbers. Simulation results, as shown in Fig. 5, provide much more information. We acquire the local response at each position along the scratching path, and also the final surface profile after full recovery of the material.

It is this type of information that gives us an opportunity to understand scratching mechanisms and scratch resistance of polymers and PBMs. Local responses at a given location can, thus, be related to the spatial structure of the chains as shown in Fig. 1b. For the future, we consider the application of focused ion beam milling [35] to PBM surfaces. We should be able to simulate milling as well as indentation (also application of an indenter, but this time a stationary rather than a moving one) by molecular dynamics.

Acknowledgments Partial financial support has been provided by the Foundation for Science and Technology (FCT), Lisbon, through projects PEst-C/CTM/LA0025/2011 and PTDC-EME-PME-108859-2008 and also by the II-VI Foundation, Bridgeville, Pennsylvania.

References

1. Jost HP (1966) The introduction of a new technology, report from the committee on tribology. Her Majesty's Stationary Office, London
2. Jost HP (1975) Economic impact of tribology. *Mech Eng* 97:26–33
3. Rabinowicz E (1965) *Friction and Wear of Materials*. Wiley, New York
4. Rabinowicz E (1995) *Friction and Wear of Materials*, 2nd edn. Wiley, New York
5. Czichos H (1978) *Tribology*. Elsevier, Amsterdam
6. Hedenqvist MS, Gedde UW (1999) Parameters affecting the determination of transport kinetics data in highly swelling polymers above T_g . *Polymer* 40:2381–2393
7. Brostow W, Deborde J-L, Jaklewicz M, Olszynski P (2003) *Tribology with emphasis on polymers: friction, scratch resistance and wear*. *J Mater Educ* 25:119–132
8. Brostow W, Kubát J (1993) Molecular dynamics simulation of stress relaxation on a triangular lattice. *Phys Rev B* 47:7659
9. Blonski S, Brostow W, Kubát J (1994) Molecular dynamics simulation of stress relaxation in metals and polymers. *Phys Rev B* 49:6494
10. Matozza A, Serralunga P, Hedenqvist MS, Gedde UW (2006) Mesoscale modelling of penetrant diffusion in computer-generated polyethylene-spherulite-like structures. *Polymer* 47:5588–5595
11. Ritums J, Neway B, Doghieri F, Bergman G, Gedde UW, Hedenqvist MS (2007) Assessing the transport properties of organic penetrants in low-density polyethylene using a “four-component” polymer free-volume model and a comparison with a semi empirical exponential model. *J Polym Sci Phys* 45:723–734
12. Matozzi A, Minelli M, Hedenqvist MS, Gedde UW (2007) Computer-built polyethylene spherulites for mesoscopic Monte Carlo simulation of penetrant diffusion: influence of crystal widening and thickening. *Polymer* 48:2453–2459
13. Matozzi A, Hedenqvist MS, Gedde UW (2007) Diffusivity of n-hexane in poly(ethylene-co-octene)s assessed by molecular dynamics simulations. *Polymer* 48:5174–5180
14. Nilsson F, Gedde UW, Hedenqvist MS (2009) Penetrant diffusion in polyethylene spherulites assessed by a novel off-lattice Monte-Carlo technique. *Eur Polym J* 45:3409–3417

15. Flory PJ (1985) Selected Works. Stanford University Press, California
16. Alder BJ, Wainwright TE (1959) Studies in molecular dynamics. I. General method. *J Chem Phys* 31:459
17. Lennard-Jones JE (1924) On the determination of molecular fields—II. From the equation of state of a gas. *Proc Royal Soc London* 106:463
18. Mie G (1903) Zur kinetischen Theorie der einatomigen Körper. *Ann Physik* 316:657
19. Simoes R, Cunha AM, Brostow W (2004) Molecular deformation mechanisms and mechanical properties of polymers simulated by molecular dynamics e-Polymers no. 067
20. Simoes R, Viana JC, Dias GR, Cunha AM (2006) Influence of the interaction potential parameters on the mechanical response of simulated semi-crystalline polymeric materials. *Mater Sci Forum* 514–516:810
21. Pisanova E, Zhandarov S (2000) Fiber-reinforced heterogeneous composites. In: Brostow W (ed) Performance of plastics, Chap. 19. Hanser, Munich
22. Nogales A, Broza G, Roslaniec Z, Schulte K, Sics I, Hsiao BS, Sanz A, Garcia-Gutierrez M, Rueda DR, Domingo C, Ezquerria TA (2004) Low percolation threshold in nanocomposites based on oxidized single wall carbon nanotubes and poly(butylene terephthalate). *Macromolecules* 37:7669–7672
23. Hao N, Böhning M, Goering H, Schönhals A (2007) Nanocomposites of polyhedral oligomeric phenethylsilsesquioxanes and poly(bisphenol A carbonate) as investigated by dielectric spectroscopy. *Macromolecules* 40:2955–2964
24. Broza G, Schulte K (2008) Melt processing and filler/matrix interphase in carbon nanotube reinforced poly(ether-ester) thermoplastic elastomer. *Polym Eng Sci* 48:2033
25. Vargas AF, Brostow W, Hagg Lobland HE, Lopez BL, Olea-Mejia O (2009) Reinforcement of polymeric latexes by in situ polymerization. *J Nanosci Nanotechnol* 9:6661–6667
26. Schönhals A, Goering H, Costa FR, Wagenknecht U, Henrich G (2009) Dielectric properties of nanocomposites based on polyethylene and layered double hydroxide. *Macromolecules* 42:4165–4174
27. Brostow W, Datashvili T, Geodakyan J, Lou J (2011) Thermal and mechanical properties of EPDM/PP thermal shock-resistant ceramic composites. *J Mater Sci* 46:2445–2455
28. Brostow W, Brozynski M, Datashvili T, Olea-Mejia O (2011) Strong thermoplastic elastomers created using nickel nanopowder. *Polym Bull* 59:1671
29. Purohit PJ, Wang D-Y, Emmerling F, Thünemann AF, Heinrich G, Schönhals A (2012) Arrangement of layered double hydroxide in a polyethylene matrix studied by a combination of complementary methods. *Polymer* 53:2245–2254
30. OpenGL—the industry standard for high performance graphics (2012) www.opengl.org/
31. Brostow W, Cassidy PE, Hagg HE, Montemartini PE (2001) Effects of fluoropolymer addition to an epoxy on scratch depth and recovery. *Mater Res Innovat* 6:7
32. Brostow W, Hagg Lobland HE, Narkis M (2006) Sliding wear, viscoelasticity and brittleness of polymers. *J Mater Res* 21:2422–2428
33. Brostow W, Hagg Lobland HE, Narkis M (2011) The concept of materials brittleness and its applications. *Polym Bull* 59:1697
34. Menard KP (2008) Dynamic mechanical analysis—an introduction, 2nd edn. CRC Press, Boca Raton
35. Brostow W, Gorman BP, Olea-Mejia O (2007) Focused ion beam milling and scanning electron microscopy characterization of polymer + metal hybrids. *Mater Lett* 61:1333–1336